## Optimization of high-performance field emission rare earth tungsten alloy cathodes\*

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The cathodes, as the electronic emission source of all kinds of electronic vacuum devices and spacecraft potential control system, its performance not only affects the overall efficiency of the equipment, but also limits the most important factors of the long life and high reliability of the system, and its emission principle mainly includes thermal emission and field emission, etc. Therefore, based on first-principles calculations using density functional theory, this study constructs atomic models of W cathode surfaces doped with different rare earth atoms. Using a  $(2 \times 2 \times 1)$  W (001) surface model, 1 ML of O atoms is absorbed on the top site of the surface, followed by doping rare earth atoms (La, Ce, Y) into the vacancy sites of the W-O lattice. The work functions of the system with rare earth atom coverages of 0.5 ML and 1 ML were calculated. Through liquid phase synthesis, plasma discharge sintering, and heat treatment, nano-scale second phase rare earth oxides(La<sub>2</sub>O<sub>3</sub>, CeO<sub>2</sub>, Y<sub>2</sub>O<sub>3</sub>, etc.)-tungsten cathodes were produced. Different ignition experiments were designed to simulate various operating conditions. The cascade arc plasma source was used for mass-loss and lifetime prediction tests on the cathode materials. After testing, Scanning Electron Microscopy and Energy Dispersive Spectrum microscopic characterizations of the cathode materials were conducted to analyze their composition, morphology, and elemental distribution. Optimization results reveal that the W-La, W-Ce, and W-Y cathodes prepared with this method exhibit excellent ablation resistance and plasma bombardment endurance at high temperatures. The nanoscale dispersion of the doped phases endows the cathode with superior electron emission properties, enhancing the overall efficiency of the system. Under plasma density of  $1.0 \times 10^{19} \, \mathrm{m}^{-3}$  and working temperature of 2000 K, the projected lifetime of rare earth tungsten alloy cathodes exceeds 2000 hours.

Keywords: Field emission cathode, Rare earth tungsten alloy, First-principles, calculations, Work function.

#### I. INTRODUCTION

As the primary and neutralizing electron source of vari-3 ous electronic vacuum devices and spacecraft potential con-4 trol systems, the performance of the cathodes not only af-5 fects the overall efficiency of the system, but is also the most 6 important factor limiting long service life and high reliabil-7 ity of the system [1, 2]. Compared with various conven-8 tional thermal cathodes, field emission cathodes have a series 9 of advantages, such as fast startup, room temperature opera-10 tion, no preheating delay, and high current density, which has 11 been applied in many fields such as electron beam lithogra-12 phy, vacuum diodes and space propulsion systems, and other 13 fields [3]. At present, in the field of vacuum devices, the com-14 monly used cathode materials are Ba-W cathodes, lanthanum 15 hexaboride(LaB<sub>6</sub>) and C12 A7 cathode materials, as well as 16 new cathode materials developed on the basis of this, and the 17 advantages of the application of different emitter materials 18 are different [4–7]. For example, in the Magneto Plasma Dy-19 namic Thruster(MPDT), the cathode, as a core component, is 20 in the center of the high-energy plasma plume, with an instantaneous ignition voltage of up to several thousand volts, an operating temperature higher than 1500K, and an extremely 23 harsh working environment, which puts forward high requirements for the cathode material and structure [8].

In order to improve the cathode working life and efficiency,

based on the traditional cathode electron emission theory and
 traditional cathode material types, this paper will carry out
 research from the perspective of cathode emitter material op timization

Atomic models were constructed with tungsten(W)-O surfaces doped with various rare earth atoms (La, Ce, Y), using first-principles calculations and density functional theory(DFT). Calculations for the work functions were conducted for the models of 0.5 ML and 1.0 ML doping levels.
The results showed that doping rare earth elements greatly
lowered the work function of the alloy cathode, improving
electron emission performance, and that 0.5 ML doping in WOlattice sites resulted in the lowest work function.

In the present study, nano-doped rare earth tungsten cathdo de materials were prepared using liquid-phase synthesis and
plasma discharge sintering techniques. A series of ignition
tests on the thruster prototypes were conducted along with
microstructural characterization experiments. Electron emission performance, ignition performance, and efficiency of a
tungsten alloy cathode doped with various elements and proportions were tested.

In addition, based on the working principle of cathode for various vacuum devices, in view of the existing experimental conditions, the long-life test of the whole machine is subject to greater constraints. Therefore, this study independently conducts a life assessment experiment for cathode, and conducts a mass-loss-life prediction of cathode through a self-developed cascade are plasma generator source. The results show that the nano rare-earth tungsten alloy cathode has bet-ter electron emission performance than the conventional cathode, in which the lanthanum oxide doped tungsten alloy with different mass fractions makes the cathode material escape work reduced significantly; after the independent life test of

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59 the cathode, the preliminary prediction of the rare-earth tung- 104 barrier of the metal. When electrons move to the metal sur-60 sten alloy cathode life reaches 2000 h.

## II. ELECTRON EMISSION PRINCIPLE OF FIELD **EMISSION CATHODE**

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The emission of cathodes used in thruster systems primar- 108 63 64 ily relies on the principles of thermionic emission and field 65 emission [9]. Thermionic emission follows the Richardson 66 equation:

$$j_0 = [AT_K^2] exp(-\phi_k/kT_K) \tag{1}$$

sion current density. A is the theoretical value of the mate-70 ature, and  $\phi_k$  represents the material's work function [10]. 71

Similar to the theoretical derivation of the thermionic emis-73 sion equation, Fowler and Nordheim developed the field 121 resent the performance: 74 emission theory for metals. They assumed the following: 75 (1) the distribution of band electrons conforms to the Fermi-76 Dirac distribution; (2) a smooth, planar metal surface is con- 122 <sup>77</sup> sidered, ignoring atomic-scale irregularities; (3) classical im-78 age forces affecting electrons are taken into account; (4) the 79 work function distribution is uniform. Under these assump-80 tions, the following equation holds:

$$j_0 = \frac{1.54 \times 10^{-6} \xi^2}{\phi_k} \exp \left[ -\frac{6.83 \times 10^7 \phi_k^{3/2}}{\xi} \theta(y_0) \right]$$
 (2)

In Equation(2),  $\xi$  represents the electric field strength, measured in V/cm. Where  $\theta(y_0)$  is a slow-variable function 84 of  $\xi$ ,

$$y_0 = \left(3.79 \times 10^{-4} \frac{\sqrt{\xi}}{\phi_k}\right) \tag{3}$$

According to the electron emission equations above, the zero-field emission current density  $j_0$  is closely related to pa-90 function of the cathode material, the more readily electrons 140 predict the physicochemical properties of a wide range of sys-93 ing temperature  $T_K$  and electric field strength  $\xi$  are directly 143 on quantum mechanics to study the properties of materials 95 the cathode. At lower temperatures, thermionically emitted 145 tion contains all the information of the computational system, of field-emitted electrons near the Fermi level [11].

100 rangement of lattice ions is interrupted at the metal-vacuum 150 teristics based on the orbital wave function, with the partiboundary, thereby disrupting the periodicity of the potential 151 cle density function to express the system base state of each 102 field [12]. The potential energy increases in a specific manner 152 physical quantity, to the electron density function represents 103 and approaches zero at infinity, forming the surface potential 153 the system energy [15, 16].

105 face and attempt to escape, they are hindered by this surface barrier, which is defined as the material's work function.

$$\phi_{\mathbf{k}} = E_{\mathbf{V}} - E_{\mathbf{F}} \tag{4}$$

All electrons attempting to escape from the metal must have energy at least equal to the Fermi level plus the value of the work function and follow a statistical distribution. Their average energy equals 32 KT, with each degree of freedom 112 contributing an average energy of 12 KT, consistent with the 113 results of kinetic molecular theory [13].

On the other hand, the cathode evaporation rate increases 115 sharply with rising cathode operating temperature. Cath-As shown in Equation (1),  $j_0$  represents the zero-field emis- 116 ode evaporation directly impacts the cathode's lifespan, grid emission, and inter-electrode insulation performance. Ideally, rial's emission constant,  $T_K$  is the cathode operating temper- 118 a cathode should have high emission capability, requiring a 119 low work function and minimal evaporation. Considering 120 these two requirements, a quality factor F can be used to rep-

$$F = \phi_{\mathbf{k}} \times 10^3 / T_{\mathbf{e}} (eV/K) \tag{5}$$

Te is the temperature (K) at which the material's vapor  $_{124}$  pressure reaches  $10^{-5}$  mmHg. To ensure thruster performance, the cathode temperature should not exceed its "vapor pressure temperature" Te [14].

Therefore, the selection and optimization of cathode emitter materials need to balance electronic emission performance with thermodynamic properties. Higher electron emission performance can enhance cathode discharge efficiency and overall thruster efficiency, while better thermodynamic properties extend the service life of the cathode under extreme operating conditions.

## III. FIRST-PRINCIPLES STUDY ON THE SURFACE WORK FUNCTION OF RARE EARTH TUNGSTEN ALLOY **CATHODE**

Quantum mechanics is an important foundation of modern rameters such as the material's work function, temperature, 138 physics and one of the greatest discoveries of the 20th cenand electric field strength. Theoretically, the lower the work 199 tury. Using quantum mechanics, it is possible to explain and within the material can overcome the surface potential barrier 141 tems and to quantitatively analyze the laws of their electronic to emit from the cathode surface. Additionally, the operat- 142 motion. The first principle is a computational method based proportional to the zero-field emission current density jo of 144 from the point of view of electron motion. The wave funcelectrons that overcome the barrier contribute negligibly to 146 which greatly limits the scope of its practical applications, the emission current, with the emission primarily consisting 147 and the establishment of the density functional theory solves 148 the problem of the complexity of the wave function. The ba-For the work function of cathode materials, the periodic ar- 149 sic idea of density functional theory is to change the charac-

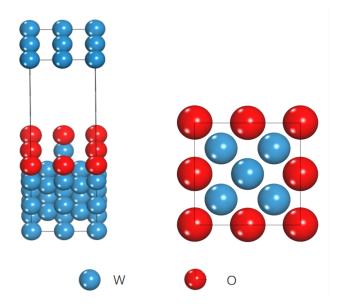


Fig. 1. Top-Site Adsorption of O Atoms on W(001) Surface.

Material Studio material simulation software incorporates 189 155 a 156 methods, which can complete the cross-scale scientific re- 191 ture, the larger the adsorption energy is, the more stable the Geometric optimization was made for tungsten alloy models 196 site shown in Figure 1 [20]. doped with various rare earth elements. Further on, the re- 197 the current density of the cathode emission [18].

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The models mainly include the following: the adsorption of O atoms and La atoms on a tungsten surface, adsorption of 173 O atoms and Ce atoms on a tungsten surface, and adsorption of O atoms and Y atoms on a tungsten surface.

The current work has used the CASTEP density functional 207 calculation module of Materials Studio, which is based on a 208 177 plane-wave basis set. Among the well-known classical al- 209 earth atoms in the W-O crystal with a coverage of 0.5 ML 178 gorithms in CASTEP, the main one is the plane-wave pseu- 210 was calculated using Equation (6). Because rare earth atoms 179 dopotential method. By moving the model of a tungsten 211 readily lose their two outermost valence electrons, transfer-180 atomic structure and geometric optimization, it was possi- 212 ring them to the inner O atoms, the electron density of the the default number of maximum steps, while the cutoff en- 214 This results in a dipole layer with a positive charge on the and for the rest of the parameters that were given, we kept 216 height. 185 them the same as the default. Under the ultrasoft pseudopo- 217 186 tential, GGA was being applied, and the functional from 218 of W-O crystals with a rare earth atom coverage of 1.0 ML, 187 Perdew-Burke-Ernzerhof was picked to describe the electron 219 meaning that rare earth atoms fully occupy the hollow sites in 188 exchange-correlated interactions [19].

O atomic layer was adsorbed on the surface of the supercell variety of three-dimensional scale simulation calculation  $^{190}$  W(2×2×1). According to the related computational literasearch from the microscopic electronic structure to the macro- 192 adsorption system is. In fact, the adsorption energies of O scopic performance prediction [17]. It is on the basis of this 193 atoms on the top site, bridge site, and hollow site of W superadvantage that the atomic-scale emission structure was mod- 194 cell are about 9.11 eV, 7.40 eV, and 8.20 eV, respectively; eled with the use of the Materials Studio in the current work. 195 this means the O atoms are preferentially adsorbed on the top

On the W-O (top-site) surface, rare earth atoms with varylaxation of surface atoms was conducted and work function 198 ing coverages were adsorbed. Similar to O atoms, rare earth was calculated under convergence conditions. The pseudopo-  $_{199}$  atoms on the  $(2 \times 2 \times 1)$  W(001)-O (top-site) surface also tential method was realized for solution of Schrödinger equa- 200 have three possible adsorption positions. Due to the larger tion, while computations of work function were performed in 201 atomic radius of rare earth elements, adsorption at the top and LDA functional and PBE-GGA functional. The work func- 202 bridge sites causes significant lattice distortion in the W lattion of the (001) crystal plane for tungsten is calculated in 203 tice. Computational results indicate that rare earth atoms are order to assess the possible effect the doped elements have on 204 more likely to adsorb at hollow sites on the W-O (top-site) 205 surface. The formula for calculating the adsorption energy of 206 rare earth atoms is as follows:

$$E_{ad} = -\frac{1}{N} (E_{La+W(001)-O(top)} - NE_{La} - E_{W(001)-O(top)})$$
(6)

As shown in Figure 2a to 2c, the work function for rare ble to find the ground state with the lowest energy. We used 213 coverage layer is lower than that of the substrate surface layer. ergy was 278.0 eV. The method of calculation was "Fine" 215 outside, raising the surface potential and reducing the barrier

Similarly, this study further calculated the work function 220 the W crystal, as shown in Figure 2d to 2f.

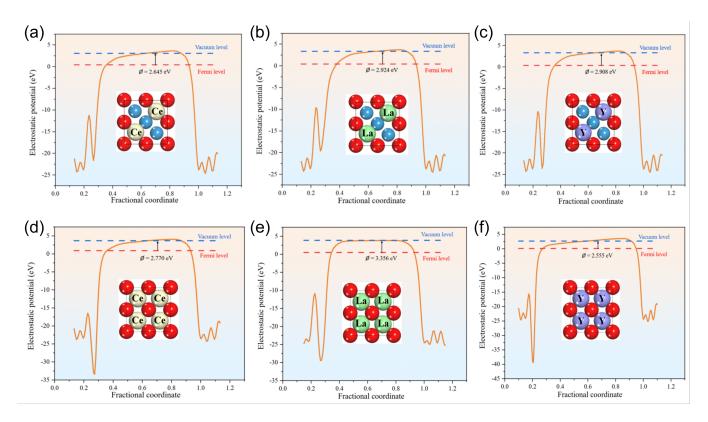


Fig. 2. W-O Crystal Surface Doped with rare earth atoms model and work function calculation.(a) Doped with 0.5 ML Ce; (b) Doped with 0.5 ML La; (c) Doped with 0.5 ML ML Y; (d) Doped with 1.0 ML Ce; (e) Doped with 1.0 ML La; (f) Doped with 1.0 ML Y.

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Table 1. Work Function of W-O (Top-Site) Doped with Different Rare Earth Atoms.

W(001)-O Top-Site Doping	Work Function (eV)
0.5 ML Ce atoms	2.645
0.5 ML La atoms	2.924
0.5 ML Y atoms	2.908
1.0 ML Ce atoms	2.770
1.0 ML La atoms	3.356
1.0 ML Y atoms	2.555
1.0 IVIL I dtollis	2.555

As marked in Table 1, the calculational results of work 246 222 function for W-O (top-site) doped with 0.5 ML and 1.0 ML of different rare earth atoms show that the large atomic radius of rare earth elements results in large lattice distortion when 224 fully occupying the hollow sites in the W crystal, and there-225 fore such a system is unstable. When the number of adsorbed 226 atoms increases above each optimal coverage, the interactions between dipoles increase gradually. Here, the middle atoms are depolarized by an electric field of adjacent dipoles. The reduction of a dipole moment increases the work function.

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into the W crystal reduces the surface work function in all optimize cathode electron emission performance by doping 258 La<sub>2</sub>O<sub>3</sub> doping concentrations [21]. 235 tungsten with rare earth elements, thereby enhancing the ef- 259 236 ficiency of magnetoplasma thrusters. Additionally, the calcu- 260 earth tungsten alloy cathode. Using liquid-phase synthesis 237 lation results show that doping 0.5 ML La or Ce atoms at the 261 and reduction, nano rare earth tungsten alloy cathode ma-

W-O (top-site) achieves the greatest reduction in work function. Due to the relatively smaller radius of Y atoms, full Y atom doping into the hollow sites of the tungsten crystal results in minimal lattice distortion, and its work function is slightly reduced compared to 0.5 ML Y doping.

## EXPERIMENTAL STUDY ON CATHODE OF RARE EARTH TUNGSTEN ALLOY

#### Synthesis and processing of rare earth tungsten alloy cathodes

For various vacuum electronics and space propulsion sys-248 tems, the primary requirements for cathodes are superior elec-249 tron emission capability and high ablation resistance to with-250 stand impacts from high-energy particles. Experimental research indicates that the electron emission performance, melt-252 ing point, and ablation resistance of rare earth tungsten alloy 253 cathodes are closely related to the chemical properties, phys-254 ical dispersion, and percentage content of the nano-doped Comparing the data in the table, doping rare earth atoms 255 phase. We synthesized W-La<sub>2</sub>O<sub>3</sub> alloy cathodes with differ-256 ent doping levels and studied their electron emission perforcases, validating the feasibility of this study's approach to 257 mance at various temperatures and voltages for different W-

Figure 3a shows the preparation process of the nano rare

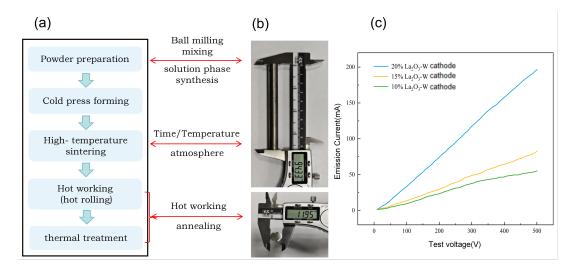


Fig. 3. Processing workflow of rare earth tungsten alloy cathodes and electron performance testing of W-La cathodes.

266 tip, resulting in lower ablation levels [22–24]. The mixed 301 take approximately 19 years [27]. powder is then processed into standard rod cathodes through 302 269 rolling), and heat treatment, as shown in Figure 3b, which 304 cathodes. This consists of a vacuum system, the power sup-270 displays a W-La cathode.

vironment of vacuum equipments and space thrusters. By 308 4a. optimizing the emitter composition, we aim to balance high 309 electron emission performance with ablation resistance for 310 the cathode changes significantly when the mass loss of the extended service life. Figure 3c shows electron emission per- 311 cathode reaches 10% - 15%. This reduces the effectiveness of 277 formance data for W cathodes doped with various mass fractions of W-La<sub>2</sub>O<sub>3</sub>. As W-La<sub>2</sub>O<sub>3</sub> content increases, the overall electron emission performance of the cathode improves sig- 314 the set operating range, and the cumulative operating time un-280 nificantly, confirming that adding low work function components to optimize emitter performance is feasible. 281

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In addition, we prepared W-Ce and W-Y rare earth tungsten 283 alloy cathodes with different doping ratios using the same processing method and made preliminary predictions of their 319 cascade arc plasma source to simulate the thruster's experi-285 properties [25, 26].

#### Independent service life experiment using cascade arc plasma source

Under conditions of existing experimental possibilities, the 326 verification of the service life of cathodes is a long-time and

262 terial precursor powder was prepared, with rare earth oxide 297 sion would have used six ion thrusters powered by nuclear 263 particles well-dispersed in the tungsten matrix. The higher 298 energy as its main propulsion; these would have to individuthe dispersion uniformity of the second-phase rare earth ox- 299 ally be rated for 83,000-hour lives. Assuming an efficient test 265 ides, the more balanced the ablation process at the cathode 300 duration of 75 percent, a 1.5-times redundancy life test would

We have developed a cascade arc plasma source to simucold pressing, high-temperature sintering, hot working (hot 303 late the real working environment of rare earth tungsten alloy 305 ply system, the superconducting magnet system, the plasma The series of rare earth tungsten alloy cathodes we de- 306 generation device, the gas supply system, the water cooling 272 signed and processed are aimed at the extreme working en- 307 system, and the Langmuir probe system, as shown in Figure

> In this paper, it is considered that the tip morphology of 312 small-hole current limitation and leads to inability to main-313 tain stable discharge when the propellant flow rate exceeds 315 der rated conditions cannot be achieved. Beyond this point, 316 the cathode is to be considered functionally degraded and at 317 the end of its service life.

> Therefore, as shown in Figure 4c to 4e, we used our custom mental environment by placing the cathode within the plasma source and setting specific plasma density and temperature parameters. Material ablation and service life predictions were conducted based on mass loss over a specified experimental duration. This study performed independent lifetime experiments and comparisons for W-La cathodes, W-Ce cathodes, and pure W cathodes [28].

The mass loss of the three experimental cathode materiexpensive process, and even more expensive are tests of life 328 als under the conditions of 3 h experimental duration with the validation conducted together with thrusters. As space mis-  $_{329}$  plasma density up to  $1.0 \times 10^{19}$  m<sup>-3</sup> measured by Langmuir sions have imposed longer lives on electric propulsion sys- 330 probe and the temperature up to 1300°C measured by the bottems, full-life ground testing has become increasingly im- 331 tom plate temperature probe is shown in Figure 4b. Based practical. For instance, the ground-tested ion thruster for 332 on parameters such as gas flow rate, input current, magnetic NASA's \*Deep Space 1\* mission lasted for 30 352 hours, 333 field strength, and plasma density, the service life of the W-296 which is more than five years. In comparison, the JIMO mis- 334 La cathode is estimated at approximately 3000 hours, and for

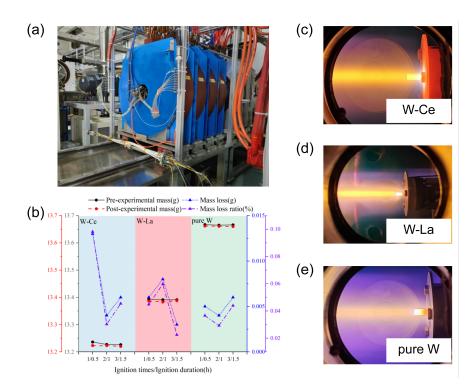


Fig. 4. Ablation life prediction experiment by cascade arc plasma generation source system.(a) Cascade arc plasma generation source system. tem(Institute of Plasma Physics, Chinese Academy of Sciences, China); (b) Test cathodes mass loss data chart; (c) W-Ce cathode test; (d) W-La cathode test; (e) Pure W cathode test.

the W-Ce cathode, about 1100 hours. This reduction is at- 361 338 duced service life. 339

340 342 mance and high ablation resistance.

### Microstructural characterization of rare earth tungsten alloy cathodes

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The melting, sputtering, and eventual deposition on the surface of the cathode material due to the W matrix are critical factors affecting the lifespan and efficiency of the cathode material. Preliminary analysis of the deposits shows no new el- 373 ements, consisting solely of tungsten oxides, which exhibit 374 ide W-La2O3, due to its low work function, allows electrons 349 valence changes at high temperatures [29]. 350

severe ablation at the tip for the rare earth-doped tungsten al- 377 ode and anode. It results in the appearance of pores, which loy cathodes. SEM images show that the mm-sized holes at 378 can be distributed on the surface of emitters in a very uneven the tip of the cathode have contracted several times due to 379 manner, and it also demonstrates that the poor work function high-temperature ablation during operation, as shown in Fig- 380 given by rare earth elements has been wildly added in order ure 5.Leading to a very irregular emitter surface that signifi- 381 to improve the total performance and efficiency of emitters. cantly influences propellant flow there at the cathode tip. This 382 Moreover, the density, size, and evenness of such pores on 358 results in more difficult ignition breakdown with the increase 383 the surface depend on many factors, including the size of the of ignition duration and frequency under the same power con- 384 second phase particle, the doping mass fraction, and the raw 360 ditions.

The W-La cathode after the experiment was further anatributed to decreased thermal resistance of the cathode matrix 362 lyzed by EDS energy spectrum, as shown in Figure 6. Among as rare earth element content increases, leading to higher mass 363 them, Figure 6a is the W-La tip deposit, and the analysis reloss under extreme operating conditions and consequently re- 364 sults show that the content of W is as high as 65.7%, followed 365 by O element, and the content of rare earth La is as low as Thus, in optimizing the performance of cathodes, it is 366 1.6%, indicating that La is doped into the W matrix in the essential to ensure both excellent electron emission perfor- 367 form of W-La<sub>2</sub>O<sub>3</sub>, which is consumed during the discharge 368 process to form electron emission. The analysis of particulate 369 matter on the cathode surface in Figure 6b shows that it is 370 the oxide of W, and the matrix of W melts and recrystallizes 371 at high temperature, changing valence and forming different 372 oxides of W.

The characterization results show that the rare earth ox-375 near the Fermi level of La to overcome the surface barrier and The nominal operation tests of the ignition demonstrated 376 emit from the material under high voltage between the cath-385 material mixing methods.

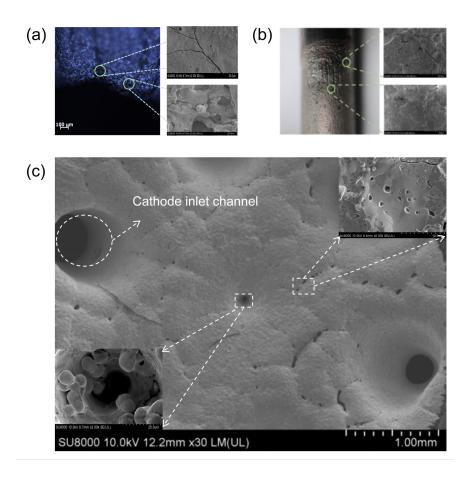


Fig. 5. Scanning Electron Microscopy (SEM) of W-La Cathode after the Experiment.(a) Tip area; (b) Transition area and normal area; (c) Tip hole area.

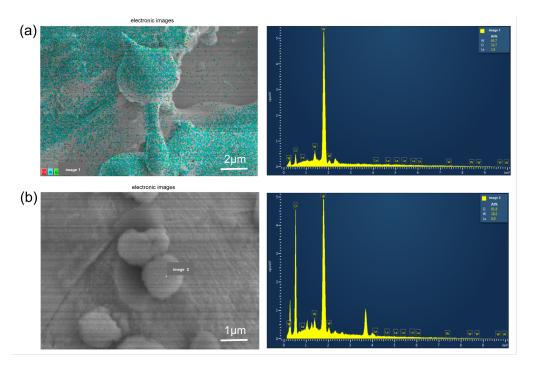


Fig. 6. Energy Dispersive Spectrum (EDS) of W-La Cathode after the Experiment.(a)Area scanning; (b)Spot scanning.

# V. RESULTS AND DISCUSSION

389 the electron emission performance and ablation resistance

390 of cathode materials against high-energy particle impacts di- 424 ble with lower ignition voltage. A cascade arc plasma source 391 rectly affect the overall efficiency, performance and life of the 425 custom's development was also employed to simulate the real 392 system and other key indicators [30]. Therefore, the future 426 working environment that tungsten alloy cathodes face. On 393 research on cathode needs to consider the doping elements 427 the basis of all the same quantity of these operational param-394 affecting the electron escape work of cathode materials, thus 428 eters, namely, the gas flow rate, input current, magnetic field dition, the ratio of doping elements and W matrix affects the 430 ered rare earth tungsten alloy cathodes were estimated. melting point of the overall alloy, that is, it affects the sput- 431 tering resistance of cathode, etc.; in addition, the preparation 432 cathodes postexperiment. The experimental results obtained 399 method of cathode also greatly affects its performance, and 433 show that the role of the rare earth doped in the tungsten alloy the optimization of the cathode preparation process enables 434 cathodes is to excite the outer valence electrons at high voltthe dopant phase to form a nano-sized in W matrix. disper- 435 age and break down the propellant to form plasma. In consion, the better the dispersion, the better the improvement ef- 496 tinuous electron emission, the consumption of the rare earth 402 fect on the cathode emission performance [31].

emission rare earth tungsten alloy cathode is investigated by 439 sistance of the cathode declines and serious ablation occurs. both simulation and experimental verification. Atomic models 440 Thus, by changing the ratio of doping atoms in the cathode, an are built with Material Studio for rare earth elements adsorbed 441 optimization of the trade-off between excellent electron emisonto the W-O (top-site) surface; the relevant surface work 442 sion performance and good resistance against ablation can be 409 functions are then calculated based on density functional the-443 achieved. ory. The work function values calculated were 2.645 eV, 444 2.924 eV, and 2.908 eV for 0.5 ML0.5 ML of Ce, La, and 445 sten alloy cathodes, we believe that high-performance field 412 Y adsorbed on the W-O top site and 2.770 eV, 3.356 eV, and 446 emission rare earth tungsten alloy cathodes will have great ap-413 2.555 eV for 1.0 ML of Ce, La, and Y, correspondingly. These 447 plication prospects in many aspects such as vacuum devices, 414 results reflect that doping of rare earth atoms effectively re- 448 space propulsion, etc., in which the excellent electron emisduces the surface potential barrier of the cathode, which con- 449 sion performance of tungsten-lanthanum cathode can greatly 416 firms that doping of rare earth atoms is an effective method to 450 improve the emission efficiency of the cathode itself. The abenhance performances of electron emission for cathodes.

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419 in the paper, W-La, W-Ce, and W-Y cathodes with different 453 focus. 420 treatment methods were prepared; a serial of ignition tests 421 have been designed and performed; the results indicate that 422 compared to pure tungsten cathodes, the operating condition 454 423 for the rare earth tungsten alloy cathode operating is more sta-

affecting the electron emission performance of cathode, in ad- 429 strength, and plasma density, the service lives for the consid-

Optical microscopy, SEM, and EDS were conducted on the 437 atoms is gradual. However, with the increase of the doping In this paper, the optimization of high performance field 438 ratio of rare earth, to a certain extent high-temperature re-

Based on the above research on different rare earth tung-<sup>451</sup> lation mechanism of rare earth tungsten alloy cathode and the Of course, based on the Material Studio simulation result, 452 development of longer-life cathode will be the next research

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